

PAUL SCHERRER INSTITUT



Matthias Krack :: Paul Scherrer Institute

# Implicit Solvent Methods and DFT+U

CP2K User Tutorial: "Computational Spectroscopy", Uni Paderborn, 28 August 2018

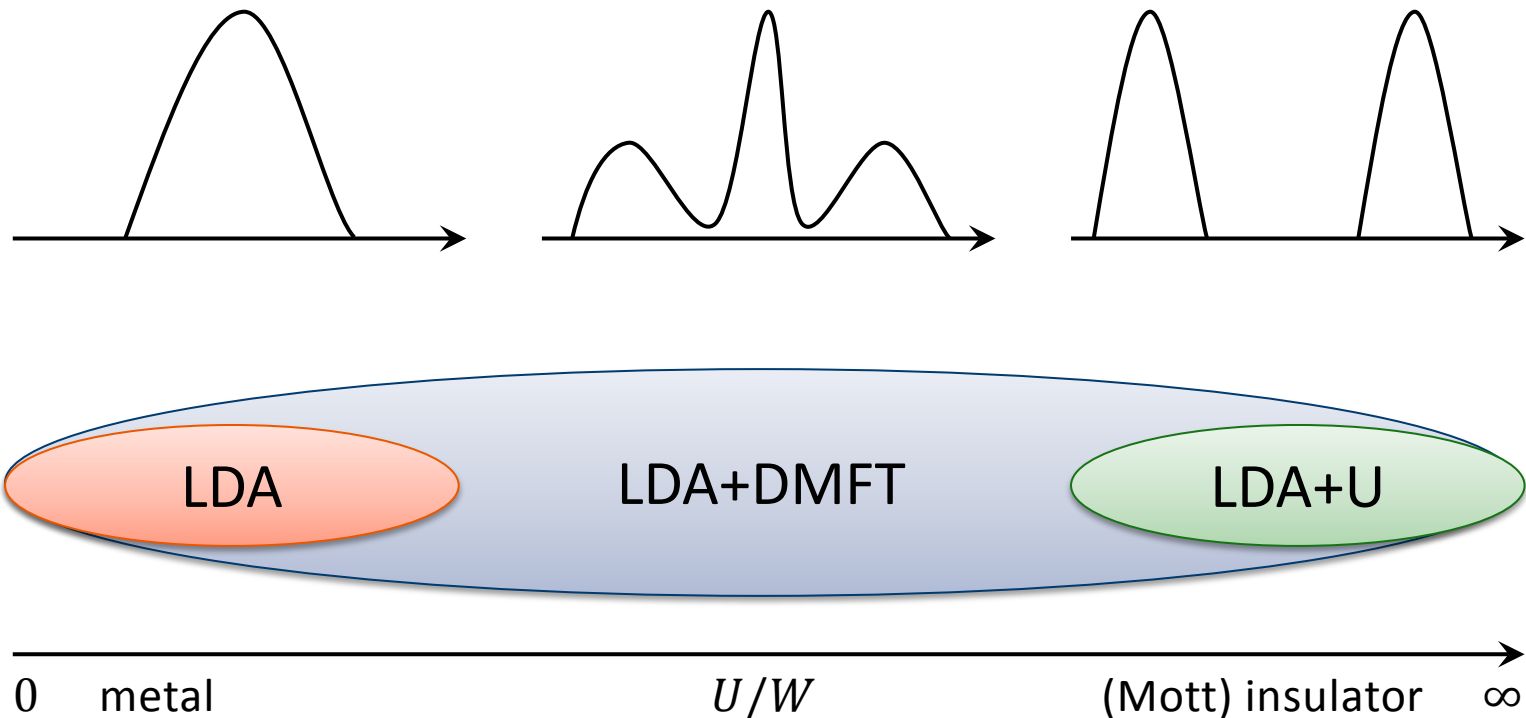
# DFT+U

# Local density approximation (LDA)

- Plain DFT (LDA, GGA) performs unexpectedly well for many solid state systems
- But:
  - LDA provides a poor description of the Coulomb interaction
  - The LDA functional is derived from a weakly correlated reference system, the jellium model (uniform or homogeneous electron gas)
  - Generalised Gradient Approximation (GGA) does not overcome these shortcomings as it is still a “local” approximation by adding only gradient information
  - Fortuitous error cancellation obscures the deficiencies
  - LDA fails badly for *strongly correlated systems* like **transition metal (TM) oxides** and **heavy fermion materials** with partially filled 4f and 5f orbitals

# Mott insulator

- Hubbard bands form as an effect of the electronic correlation:
  - LDA bands split into two sets of bands separated by a local Coulomb repulsion  $U$  and band width  $W$  → Mott insulator



# Strongly correlated systems

- Though it is possible to calculate the Hubbard parameter  $U$  using linear response theory\*, it is usually treated as an empirical fitting parameter
- Fixed  $U$  parameter does not adapt to the chemical environment (e.g. change of oxidation state)
- Occurrence of meta-stable states for heavy fermion materials like  $\text{CeO}_2$ ,  $\text{UO}_2$  etc.  $\Rightarrow$  major obstacle technically
- On-going development, e.g. dynamical mean field theory (LDA+DMFT)
  - However, such methods are computationally very expensive
  - Accessible model system sizes are quite limited currently ( $\approx 30$  atoms per cell)
  - Analytical energy gradients, i.e. atomic forces, are not straightforwardly available

\*Cococcioni and Gironcoli, Phys. Rev. B **71**, 035105

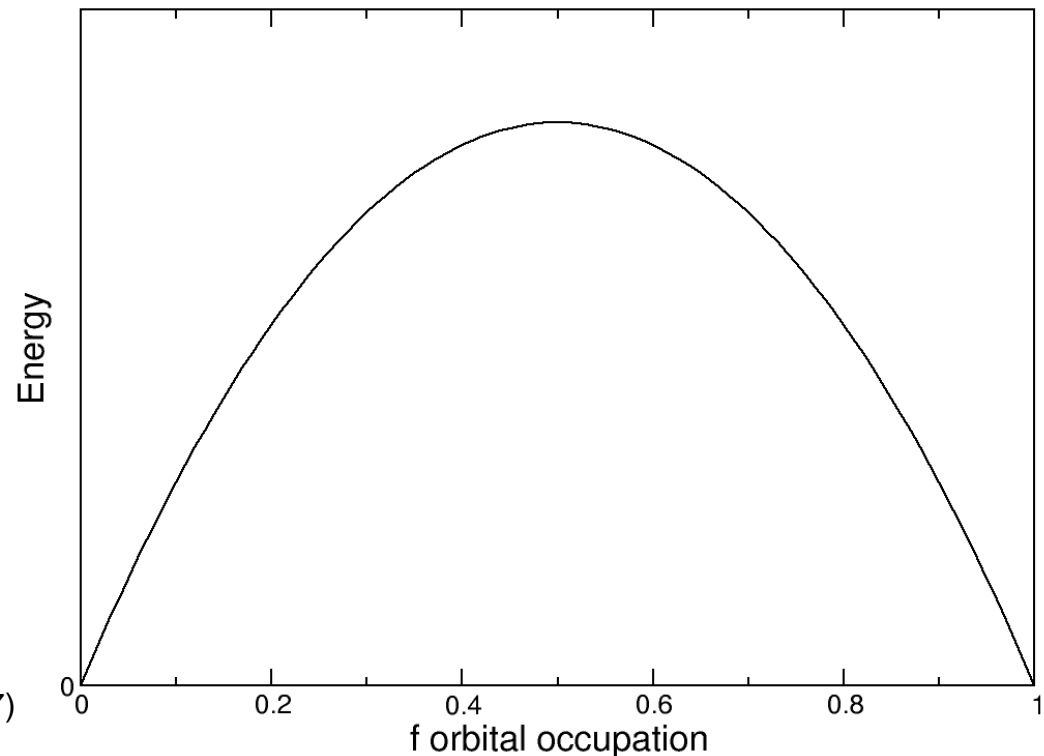
- Hubbard correction term is added to the DFT energy functional as an atomic on-site term (V. I. Anisimov et al., Phys. Rev. B **44**, 943 (1991))

- Explicit correction that acts as an energy penalty function, e.g. Dudarev\*:

$$E_U = \frac{U_{\text{eff}}}{2} \sum_{\sigma, I} \text{Tr}[\mathbf{n}^{\sigma, I} (\mathbf{1} - \mathbf{n}^{\sigma, I})]$$
 with atom  $I$  and spin  $\sigma$

- Computationally cheap
- Right physical effect
- Rotationally invariant
- Effective  $U$  parameter:
- $U_{\text{eff}} = U - J$

\*Dudarev et al., Phil. Mag. B **75**, 613 (1997)



- Orbital based methods:

- Chirgwin and Coulson:  $\mathbf{P} = \frac{1}{2}(\mathbf{D}\mathbf{S} + \mathbf{S}\mathbf{D})$

- Mulliken:  $\mathbf{P} = \mathbf{D}\mathbf{S}$

- Löwdin:  $\mathbf{P} = \mathbf{S}^{1/2}\mathbf{D}\mathbf{S}^{1/2}$

with density matrix  $\mathbf{D}$ , overlap matrix  $\mathbf{S}$ , and occupation matrix  $\mathbf{P}$

- Partitioning of the electronic density:

- Bader charges: Gradient of the electronic charge density (zero-flux surface)

- Hirshfeld charges: Difference to unrelaxed atomic densities (pro-density)

- Potential derived charges methods

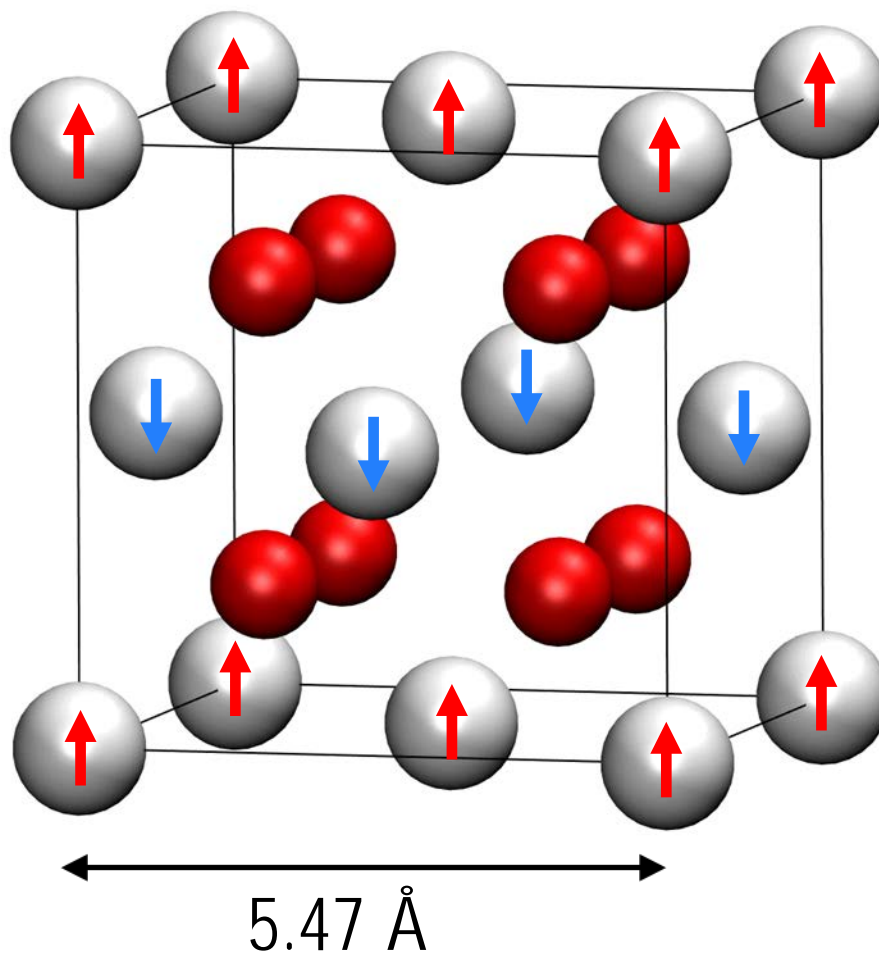
- Moment derived charges

# Tackling the meta-stable state problem

- Occupation matrix control (OMC)  
[B. Dorado et al., Phys. Rev. B \*\*79\*\*, 235125 \(2009\)](#)
- U ramping method  
[B. Meredig et al., Phys. Rev. B \*\*82\*\*, 195128 \(2010\)](#)
- Quasi-annealing (QA) method  
[H. Y. Geng et al., Phys. Rev. B \*\*82\*\*, 094106 \(2010\)](#)
- Controlled symmetry reduction (CSR) method  
[D. Gryaznov et al., Phys. Chem. Chem. Phys. \*\*14\*\*, 4482 \(2012\)](#)
- f occupation smearing and U ramping (FOUR)  
[J. Rabone and M. Krack, Comput. Mat. Sci. \*\*71\*\*, 157 \(2013\)](#)
- Local electronic minima inhibition by averaging occupations (LEMIAO)  
[J. Rabone and M. Krack, Comput. Mat. Sci. \*\*71\*\*, 157 \(2013\)](#)



# Example: Uranium dioxide (UO<sub>2</sub>)

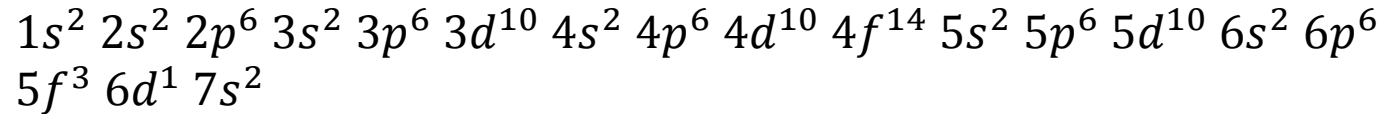


- Antiferromagnetic ground state below 30.4 K:

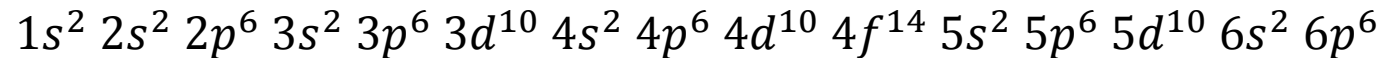
```
&FORCE_EVAL
METHOD Quickstep
STRESS_TENSOR analytical
&DFT
  CHARGE 0
  LSD
  MULTIPLICITY 1
  PLUS_U_METHOD Mulliken
  . . .
&END DFT
. . .
&END FORCE_EVAL
```

# Example: Uranium dioxide (UO<sub>2</sub>)

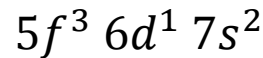
- All-electron configuration of uranium: 92 electrons



- 86 core electrons:

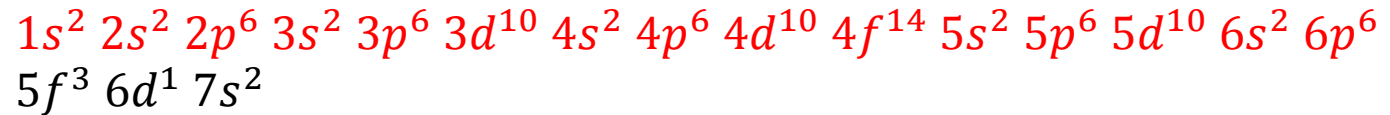


- 6 valence electrons:



# Example: Uranium dioxide (UO<sub>2</sub>)

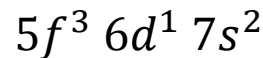
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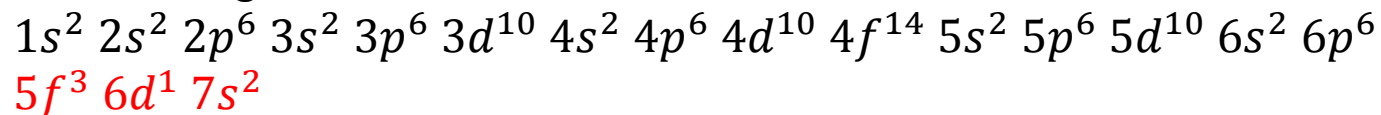


- 6 valence electrons:

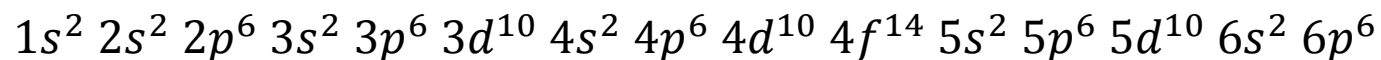


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- All-electron configuration of uranium: 92 electrons



- 86 core electrons:

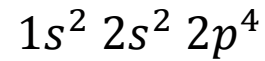


- 6 valence electrons:



# Example: Uranium dioxide (UO<sub>2</sub>)

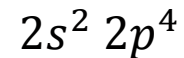
- All-electron configuration of oxygen: 8 electrons



- 2 core electrons:

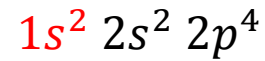


- 6 valence electrons:



# Example: Uranium dioxide (UO<sub>2</sub>)

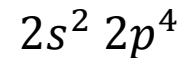
- All-electron configuration of oxygen: 8 electrons



- 2 core electrons:



- 6 valence electrons:



# Example: Uranium dioxide (UO<sub>2</sub>)

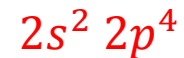
- All-electron configuration of oxygen: 8 electrons



- 2 core electrons:



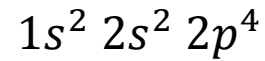
- 6 valence electrons:



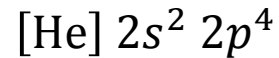


# Example: Uranium dioxide (UO<sub>2</sub>)

- All-electron configuration of oxygen: 8 electrons

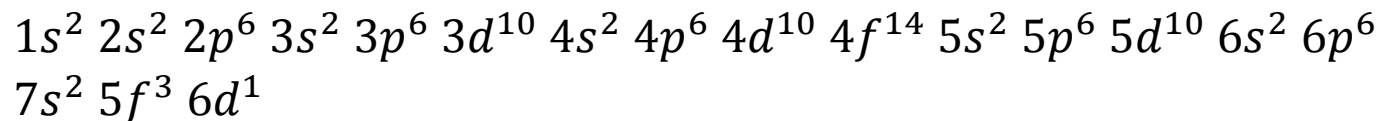


- Pseudo atom configuration of oxygen: 2 + 6 electrons

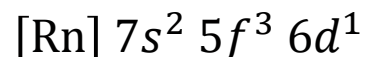


## Example: Uranium dioxide (UO<sub>2</sub>)

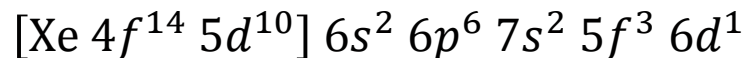
- All-electron configuration of uranium: 92 electrons



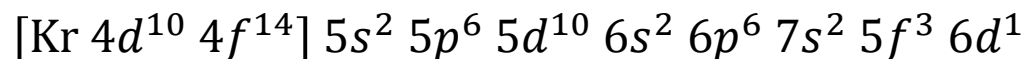
- Large-core pseudo atom configuration of uranium: 86 + 6 electrons



- Medium-core pseudo atom configuration of uranium: 78 + 14 electrons



- Small-core pseudo atom configuration of uranium: 60 + 32 electrons



# DFT+U section in the atomic kind section

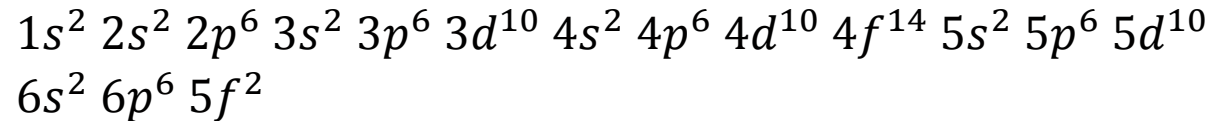
```
&KIND 0  
  BASIS_SET DZVP-MOLOPT-SR-GTH-q6  
  POTENTIAL GTH-PBE-q6  
  &BS  
    . . .  
  &END BS  
  ! Not needed for 0  
&END KIND
```

# DFT+U section in the atomic kind section

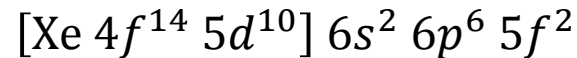
```
&KIND Ua
BASIS_SET DZVP-MOLOPT-GTH-q14
ELEMENT U
POTENTIAL GTH-PBE-q14
&BS
. . .
&END BS
&DFT_PLUS_U on
L 3
U_MINUS_J [eV] 2.00
&ENFORCE_OCCUPATION on/off
. . .
&END ENFORCE_OCCUPATION
&END DFT_PLUS_U
&END KIND
```

# Example: Uranium dioxide (UO<sub>2</sub>)

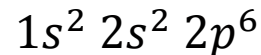
- Electronic configuration of U<sup>4+</sup>:



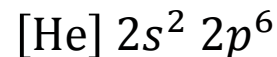
- Medium-core pseudo atom configuration of U<sup>4+</sup>:



- Electronic configuration of O<sup>2-</sup>:

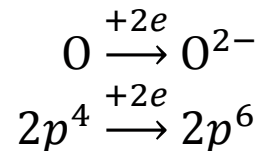


- Pseudo atom configuration of O<sup>2-</sup>:



# BS section: Initial atomic orbital occupations

- Set up the (on-site) atomic orbital occupations for  $O^{2-}$ :



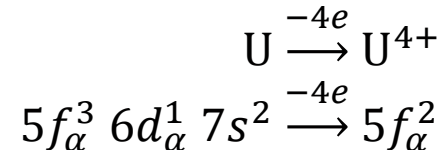
- BS section input is not processed quite intuitively:

```

&BS
&ALPHA
  N    2 ! 2
  L    1 ! p
  NEL +2 ! (4 + 2)/2 = 3 alpha 2p electrons
&END ALPHA
&BETA
  N    2 ! 2
  L    1 ! p
  NEL +2 ! (4 + 2)/2 = 3 beta 2p electrons
&END BETA
&END BS
  
```

# BS section: Initial atomic orbital occupations

- Set up the (on-site) atomic orbital occupations for  $U^{4+}$ :



- On-site triplet state for  $U^{4+}$  for the spin-up (alpha) uranium atoms **Ua** and swap **&ALPHA** and **&BETA** sections for the spin-down (beta) **Ub** kind (not shown):

**&BS**

**&ALPHA**

N      5      6      7

L      3      2      0

NEL    +1    -1    -2 ! (3 + 1)/2 = 2 alpha 5f electrons

**&END ALPHA**

**&BETA**

N      5      6      7

L      3      2      0

NEL    -3    -1    -2 ! remove all beta valence electrons

**&END BETA**

**&END BS**

# Example: Uranium dioxide (UO<sub>2</sub>)

- Two alpha electrons in seven 5f orbitals:  $\binom{7}{2} = 21$  combinations

	$f_{-3}$	$f_{-2}$	$f_{-1}$	$f_0$	$f_{+1}$	$f_{+2}$	$f_{+3}$
1	↑	↑					
2	↑		↑				
3	↑			↑			
4	↑				↑		
5	↑					↑	
6							↑
7		↑	↑				
8		↑		↑			
9		↑			↑		
10		↑				↑	

	$f_{-3}$	$f_{-2}$	$f_{-1}$	$f_0$	$f_{+1}$	$f_{+2}$	$f_{+3}$
11		↑					↑
12			↑	↑			
13			↑		↑		
14			↑			↑	
15			↑				↑
16				↑	↑		
17				↑		↑	
18				↑			↑
19					↑	↑	
20					↑		↑
21						↑	↑





# Print DFT+U specific information

- Print occupation of each atom with a  $U$  value greater zero:

```
&GLOBAL
```

```
  PRINT_LEVEL medium
```

```
  . . .
```

```
&END GLOBAL
```

```
&FORCE_EVAL
```

```
  . . .
```

```
  &DFT
```

```
    . . .
```

```
    &PRINT
```

```
      &PLUS_U on
```

```
      &EACH
```

```
        QS_SCF 1
```

```
      &END EACH
```

```
    &END PLUS_U
```

```
  &END PRINT
```

```
    . . .
```

```
  &END DFT
```

```
  . . .
```

```
&END FORCE_EVAL
```

# Print DFT+U specific information

- Use a tiny  $U$  value to trigger printout for the  $U = 0$  case:

```
&DFT_PLUS_U
```

```
. . .
```

```
U_MINUS_J [eV] 2.00E-20
```

```
. . .
```

```
&END DFT_PLUS_U
```

- A specific (initial) orbital occupation can be enforced:

```
&DFT_PLUS_U
```

```
. . .
```

```
&ENFORCE_OCCUPATION on/off
```

```
EPS_SCF 1.0E-5 ! Enforce until a certain SCF convergence is reached
```

```
MAX_SCF 20 ! Enforce occupation for first 20 SCF iterations
```

```
ORBITALS -3 -2 -1 +0 +1 +2 +3 ! Smear f electrons over all f orbitals
```

```
SMEAR on
```

```
&END ENFORCE_OCCUPATION
```

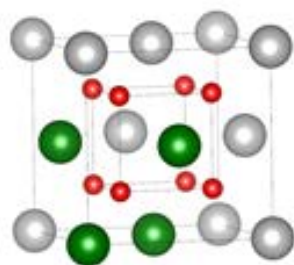
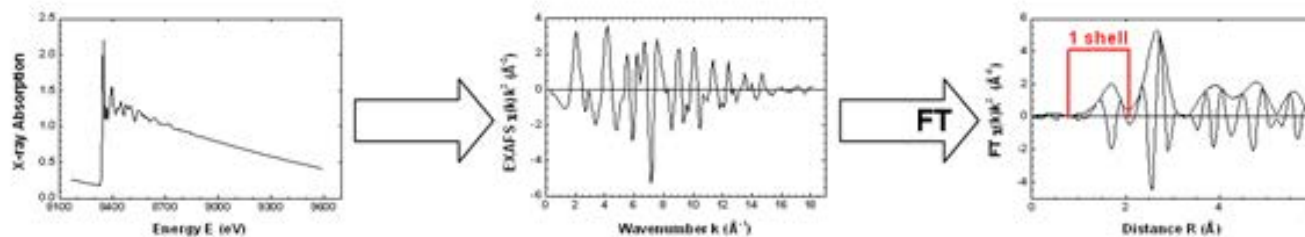
```
. . .
```

```
&END DFT_PLUS_U
```

- Alternatively, define and enforce a specific orbital occupation pattern, e.g.

```
ORBITALS -3 -1
```

# Example: XAFS spectra simulation



Crystal structure model

atoms.inp

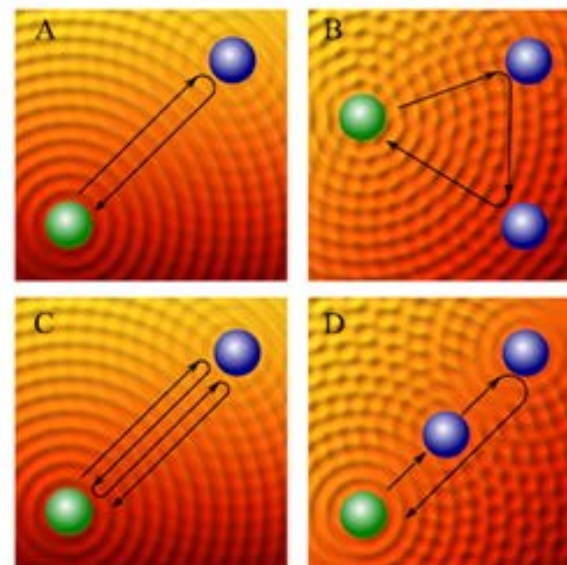
ATOM

feff.inp

FEFF8

$\chi_{\text{FEFF}}(k)$

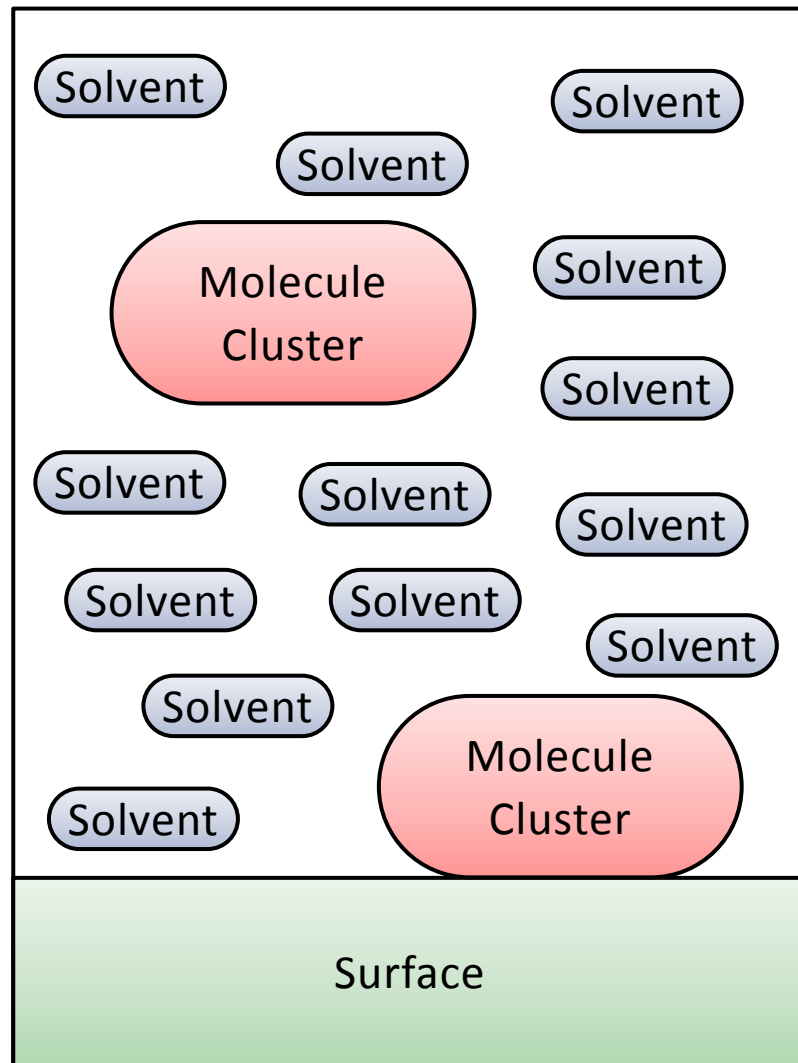
Compare  $\chi(k)$  phase



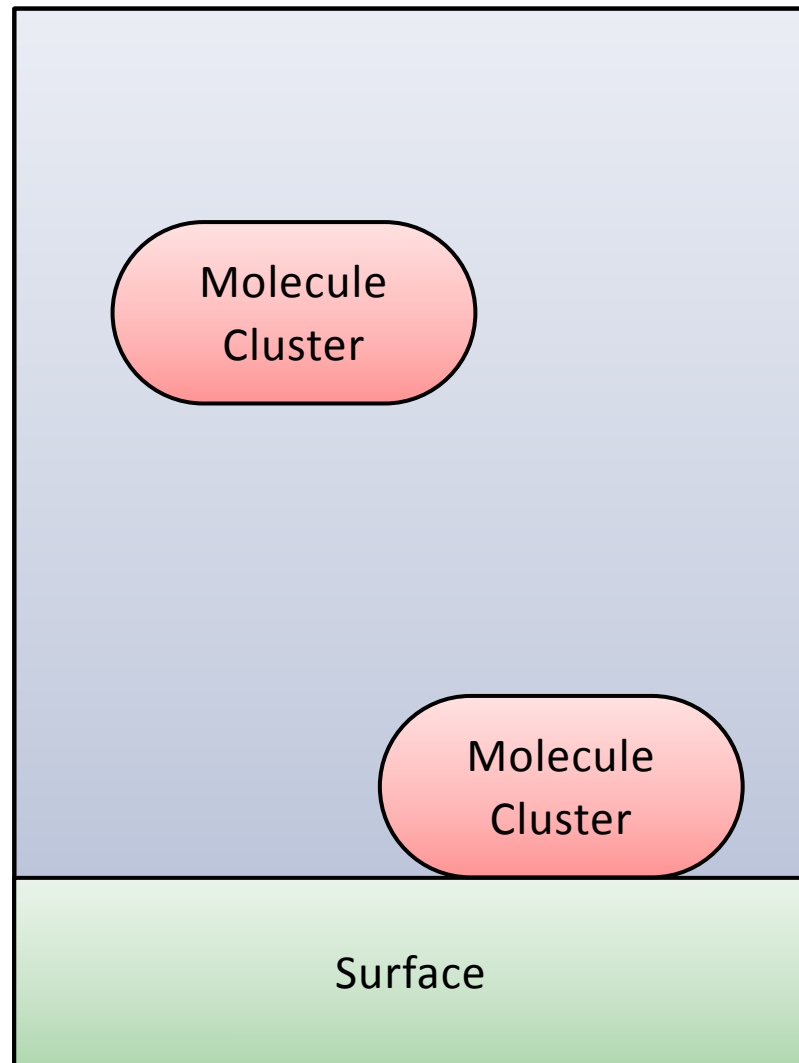
# Implicit Solvent Methods

# Modelling of solvation effects

- Vacuum
- Gas phase
- Explicit solvent with solute



- Vacuum
- Gas phase
- Explicit solvent with solute
- Implicit solvent with solute



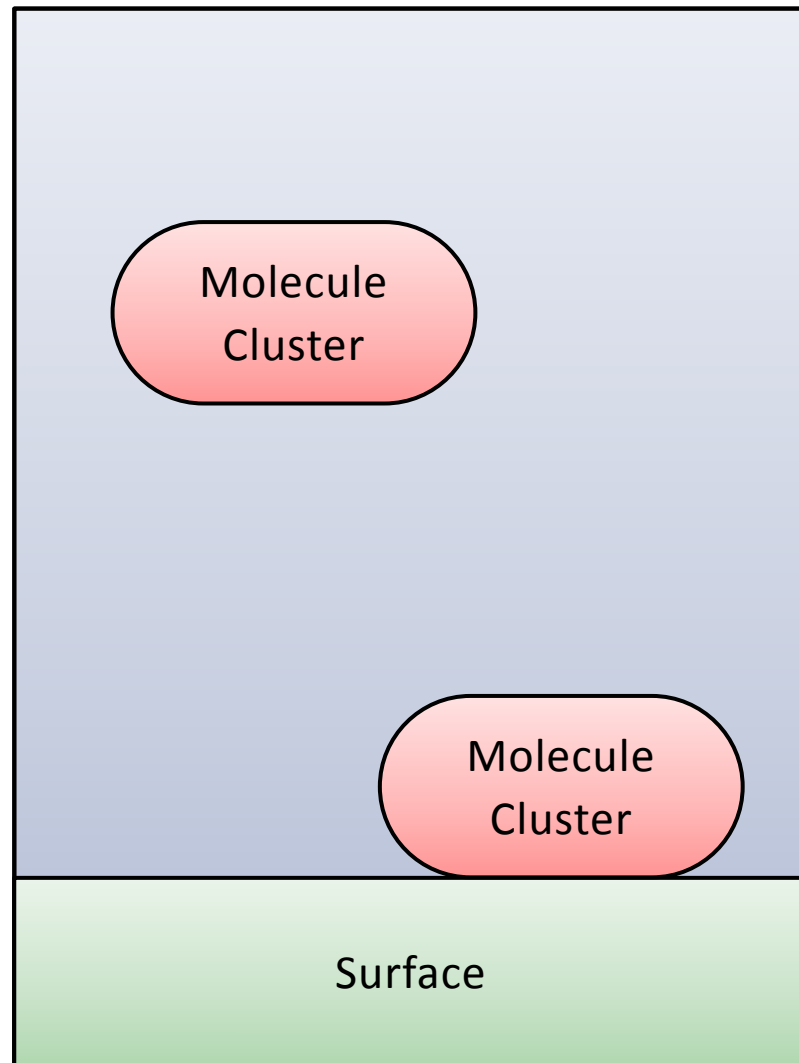
# Continuum solvation models

- Polarisable continuum model (PCM, Tomasi et al.)
- Conductor-like screening model (COSMO, Klamt et al.)
- Smooth dielectric models are needed for molecular dynamics (MD) simulations



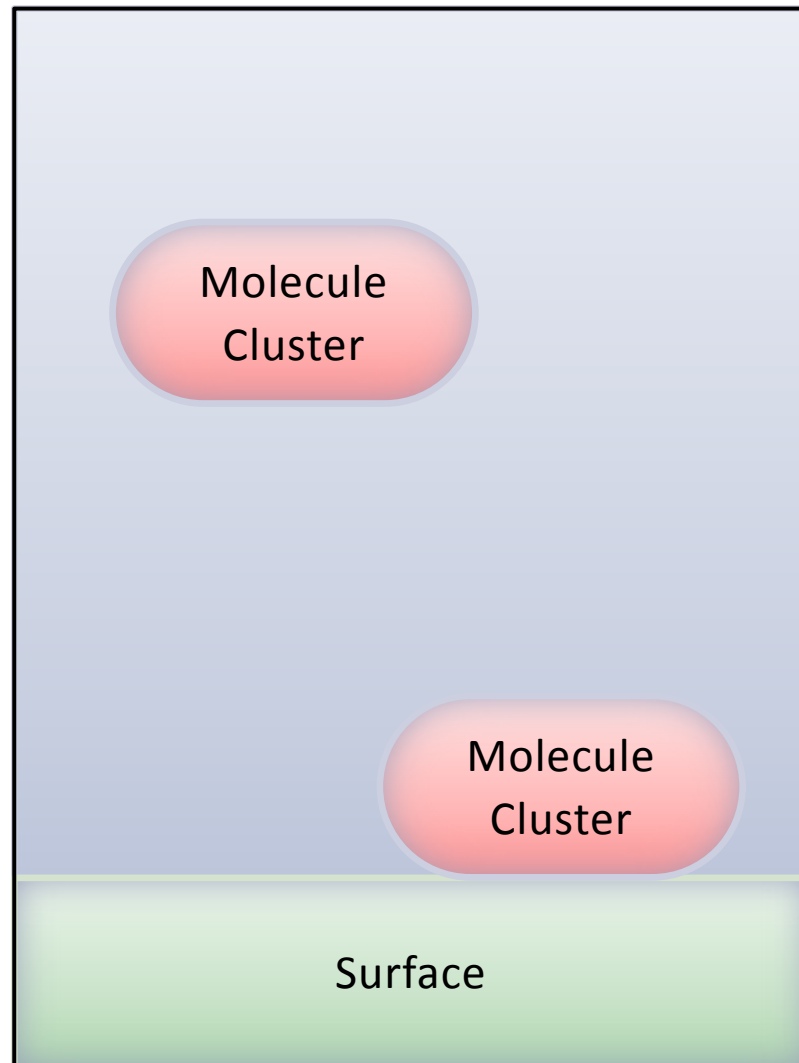
# Modelling of solvation effects

- Vacuum
- Gas phase
- Explicit solvent with solute
- Implicit solvent with solute



# Modelling of solvation effects

- Vacuum
- Gas phase
- Explicit solvent with solute
- Implicit solvent with solute
- Smoothed solute-solvent transition



# Pros and Cons of an implicit solvent approach

- Explicit solute-solvent interactions are missing
  - Pros:
    - Shortcomings of standard DFT in the description of van der Waals and hydrogen bonds w.r.t. the solvent do not matter
    - Shorter sampling times needed to obtain meaningful thermodynamic averages
  - Cons:
    - No detailed description of hydrogen bonds (network)
    - Potential (chemical) reactions between solvent and solute are excluded a priori
  
- No explicit solvent atoms
  - Pros:
    - Significant reduction in problem size especially for localised basis set methods
    - More efficient sampling
  - Cons:
    - Additional (nested) convergence cycle for polarisation charge required
    - No information about solvation shells (detailed feedback of the solvent)
    - Potentially more noisy forces due to finite differences approach

# Smoothed dielectric function

- Dielectric as a smoothed self-consistent function of the electronic density:

$$\epsilon(\mathbf{r}) \equiv \epsilon[\rho^{\text{elec}}(\mathbf{r})] = \begin{cases} 1 & \text{large } \rho^{\text{elec}} \\ \epsilon_0 & \rho^{\text{elec}} \rightarrow 0 \end{cases}$$

- J.-L. Fattebert and F. Gygi, J. Comput. Chem. **23**, 662 (2002)
- J.-L. Fattebert and F. Gygi, Int. J. Quantum Chem. **93**, 139 (2003)
- O. Andreussi et al., J. Chem. Phys. **136**, 064102 (2012)

- Adding a dielectric medium to the system

$$\epsilon(\mathbf{r}) \equiv \epsilon[\rho^{\text{elec}}(\mathbf{r})]$$

$$\nabla^2 \phi^{\text{tot}}(\mathbf{r}) = -4\pi \rho^{\text{solute}}(\mathbf{r}) \quad \text{with } \rho^{\text{solute}}(\mathbf{r}) = \rho^{\text{ions}}(\mathbf{r}) + \rho^{\text{elec}}(\mathbf{r})$$

$$\nabla \cdot \epsilon[\rho^{\text{elec}}(\mathbf{r})] \phi^{\text{tot}}(\mathbf{r}) = -4\pi \rho^{\text{solute}}(\mathbf{r})$$

$$\nabla \cdot \mathbf{E}(\mathbf{r}) = 4\pi \rho^{\text{solute}}(\mathbf{r}) - 4\pi \cdot \mathbf{P}(\mathbf{r})$$

$$\rho^{\text{pol}}(\mathbf{r}) = -\nabla \cdot \mathbf{P}(\mathbf{r}) = \nabla \cdot \left( \frac{\epsilon(\rho^{\text{elec}}(\mathbf{r})) - 1}{4\pi} \nabla \phi^{\text{tot}}(\mathbf{r}) \right)$$

- Finally a vacuum-like Poisson problem is recovered

$$\nabla^2 \phi^{\text{tot}}(\mathbf{r}) = -4\pi (\rho^{\text{solute}}(\mathbf{r}) + \rho^{\text{pol}}(\mathbf{r}))$$

# Self-consistent Continuum Solvation (SCCS)

- Vacuum-like Poisson problem is recovered

$$\nabla^2 \phi^{\text{tot}}(\mathbf{r}) = -4\pi(\rho^{\text{solute}}(\mathbf{r}) + \rho^{\text{pol}}(\mathbf{r}))$$

- Energy term

$$E^{\text{el}} = E^{\text{solute}} + E^{\text{pol}}$$

# Solvation free energy

- Solvation free energy

$$\Delta G^{\text{sol}} = \Delta G^{\text{el}} + G^{\text{rep}} + G^{\text{dis}} + G^{\text{cav}} + \Delta G^{\text{tm}} + P\Delta V$$

- Electrostatic contribution:

$$\Delta G^{\text{el}} = G^{\text{el}} - G^0$$

with the energy  $G^0$  of the solute in vacuum

- Repulsion term\*:

$$G^{\text{rep}} = \alpha S$$

where  $S$  is the (quantum) surface of the cavity

- Dispersion term\*:

$$G^{\text{dis}} = \beta V$$

where  $V$  is the (quantum) volume of the cavity

\*Scherlis et al., J. Chem. Phys. 124, 074103 (2006)

# Solvation free energy

- Cavitation term\*:

$$G^{\text{cav}} = \gamma S$$

where  $S$  is the (quantum) surface of the cavity

- Thermal motion term  $G^{\text{tm}}$  and the volume change term  $P\Delta V$  are often ignored
- Collecting all terms results in an approximation for the solvation free energy

$$\Delta G^{\text{sol}} \approx \Delta G^{\text{el}}(\epsilon_0, \rho_{\text{min}}, \rho_{\text{max}}) + (\alpha + \gamma) S + \beta V$$

- Quantum volume  $V$  and surface  $S$ :
  - M. Cococcioni et al., Phys. Rev. Lett. **94**, 145501 (2005)

\*Scherlis et al., J. Chem. Phys. 124, 074103 (2006)



# SCCS input section

```
&SCCS on/off
  ALPHA [N*m^-1] 0.0
  BETA [kbar] 0.0
  DELTA_RHO 2.0E-5
  DERIVATIVE_METHOD cd3/cd5/cd7/fft
  DIELECTRIC_CONSTANT 78.36
  EPS_SCCS 1.0E-6
  GAMMA [mN/m] 0.0
@IF ${OT}
  EPS_SCF 0.03
@ENDIF
@IF ${TD}
  EPS_SCF 0.3
@ENDIF
  MAX_ITER 100
  METHOD Andreussi/Fattebert-Gygi
  MIXING 0.6
  . . .
&END SCCS
```

- Fattebert-Gygi

$$\epsilon[\rho^{\text{elec}}(\mathbf{r})] = 1 + \frac{\epsilon_0 - 1}{2} \left( 1 + \frac{1 - (\rho^{\text{elec}}/\rho_0)^{2\beta}}{1 + (\rho^{\text{elec}}/\rho_0)^{2\beta}} \right)$$

- Andreussi et al.

$$\epsilon[\rho^{\text{elec}}(\mathbf{r})] = \begin{cases} 1 & \rho^{\text{elec}} > \rho_{\text{max}} \\ \exp(t(\ln \rho^{\text{elec}})) & \rho_{\text{min}} < \rho^{\text{elec}} < \rho_{\text{max}} \\ \epsilon_0 & \rho^{\text{elec}} < \rho_{\text{min}} \end{cases}$$

$$t(x) = \frac{\ln \epsilon_0}{2\pi} \left[ 2\pi \frac{\ln \rho_{\text{max}} - x}{\ln \rho_{\text{max}} - \ln \rho_{\text{min}}} - \sin \left( 2\pi \frac{\ln \rho_{\text{max}} - x}{\ln \rho_{\text{max}} - \ln \rho_{\text{min}}} \right) \right]$$

```
&SCCS on/off
```

```
. . .
```

```
DIELECTRIC_CONSTANT 78.36
```

```
METHOD Andreussi/Fattebert-Gygi
```

```
. . .
```

```
&ANDREUSSI
```

```
  RHO_MAX 0.001
```

```
  RHO_MIN 0.0001
```

```
&END ANDREUSSI
```

```
. . .
```

```
&FATTEBERT-GYGI
```

```
  BETA 1.3
```

```
  RHO_ZERO 0.0004
```

```
&END FATTEBERT-GYGI
```

```
. . .
```

```
&END SCCS
```

# Questions or comments?

